



## The atomic and electronic structure of amorphous BP<sub>4</sub>

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### ABSTRACT

First-principles simulation and PACVD experiment are used to reveal the atomic and electronic structure of BP<sub>4</sub>. With melting–quenching technique in molecule dynamics, for the first time we obtain the microscopic atomic picture for the amorphous BP<sub>4</sub>. Here, boron essentially retains a coordination number of 4 and *sp*<sup>3</sup> electronic bonding whereas phosphorus has a lower coordination number of 3.2 and coexistence of *sp*<sup>3</sup> and *p* electronic bonding. The cohesive energy of the present amorphous BP<sub>4</sub> model is up to −5.32 eV/atom which is 1.6 times stronger than −3.37 eV/atom of the usual protected *c*-ZnS.

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### 1. Introduction

Boron phosphide (BP) is a III–V semiconductors with zincblende (ZB) structure [1]. BP exhibits high synthesis temperature of 1130 °C and high melting point of 3000 °C [2,3]. It has attracted great interests on the mechanical properties due to its high hardness. More recently, many efforts have been made to fabricate BP thin film, since it can be used as a protective coating for high velocity infrared transmitting substrates, for example ZnS and Ge [4,5]. Different techniques, such as reactive sputtering (RS), metal organic chemical vapor deposition (MOCVD) and plasma assisted chemical vapor deposition (PACVD), have been reported to prepare BP thin film [6,7]. Among these methods, PACVD is the most popular because it can deposit BP thin film on the surface of optical device with large area and special geometry at low temperature (<500 °C) [5,8]. The BP thin film produced by PACVD is amorphous and non-stoichiometric, BP<sub>x</sub>, where *x* was about 4–8. It is widely accepted that the microstructure of the amorphous BP<sub>x</sub> is sustained by the random network of P polymer, which is pinned by B atoms [4,9]. While the microscopic atomic structure of the amorphous BP<sub>x</sub> thin film is still too difficult to be observed from experiment.

The theoretical work on ZB BP have been extensively performed, mostly carried out by Wentzcovitch et al. [10,11], Alves et al. [12] and Ferhat et al. [13,14]. Specifically, the structural and electronic properties of ZB BP have recently, calculated by Ferhat [15–17], Zaoui [15–17] and other research group [18] by using

first-principles methods based on the density functional theory (DFT). However, there is no investigation on the microstructure of the amorphous BP<sub>x</sub> thin film by theoretical calculation. First-principles MD simulation can provide unique insights on the structural evolution and especially obtaining amorphous structure [19]. Herein, we use first-principles MD simulation combined with PACVD experiment to reveal the microscopic atomic structure of amorphous BP<sub>x</sub> thin film.

### 2. Methods and computational details

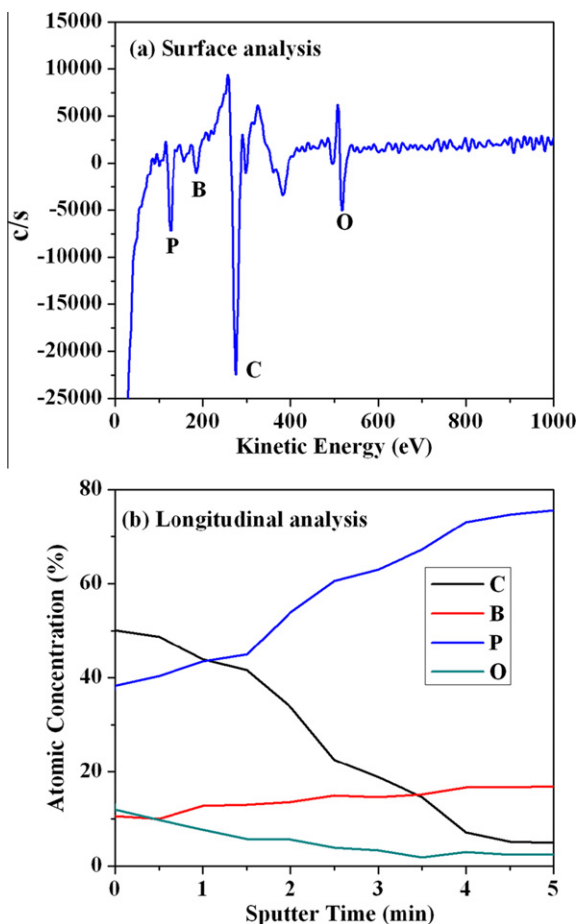
We use the first-principles MD and structural optimization to reveal the atomic image and electronic structure of amorphous BP<sub>4</sub> thin film. In the MD simulation, the melting–quenching (M–Q) method, which will be detailed later, is used to obtain the model of amorphous BP<sub>4</sub>. The density functional theory with the generalized gradient approximation [20] is employed, as implemented in the VASP code [21,22]. The electron–ion interaction is described by the frozen-core all-electron projector augmented wave (PAW) method [23]. An energy cutoff of 350 eV is used for the plane wave expansion.  $\Gamma$  point is used in the Brillouin zone sampling. In the MD simulation, we used the canonical NVT ensemble, in which the Nosé–Hoover thermostat is used to control the temperature [24,25].

### 3. Results and discussion

We prepare BP<sub>x</sub> thin film on the substrate of ZnS at 450 °C by PACVD and confirm the *x* value by auger electron spectroscopy (AES). Fig. 1a and b are the surface and longitudinal analysis of BP<sub>x</sub> thin film by AES. It can be seen that, apart from B and P atoms, O and C atoms also detected on the surface. With the increasing of sputtering time, the concentration of O and C atoms decreases, while the concentration of B and P atoms increases. When the

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**Fig. 1.** Surface and longitudinal analysis of BP film by Auger electron spectroscopy (AES). (a) Surface analysis and (b) longitudinal analysis.

sputtering time is over 4 min, the number of O and C atoms is much lower than that of B and P atoms and the ratio of P and B atoms reaches equilibrium, about 4. We also obtained this results by XPS analysis [26]. To predigest the simulation, we ignore the existence of O and C atoms. Therefore, the  $x$  value of  $BP_x$  is 4 ( $BP_4$ ).

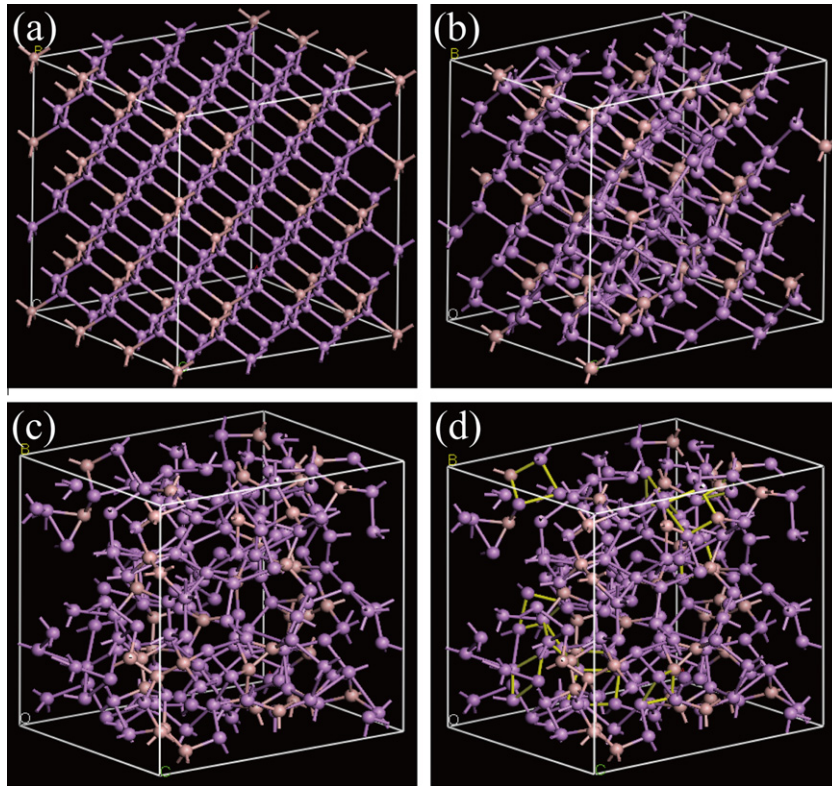
First, for the convenience of model establishment, the ZB structure is used and a supercell of  $BP_4$  with 216 atoms (43 B, 173 P) is constructed as presented in Fig. 2a. Second, to estimate the density close to the one of  $BP_4$ , structural optimization of the starting ZB  $BP_4$  is carried out. Finally, we obtain the density of the relaxed  $BP_4$  is  $0.0641 \text{ atom}/\text{\AA}^3$ , which is 25.1% lower than that of ideal ZB  $BP_4$  ( $0.0856 \text{ atom}/\text{\AA}^3$ ). In fact, the covalent radius of P (1 Å) is bigger than that of B (0.85 Å), so when the concentration of P increases the density of  $BP_4$  would obviously decreases. Before the MD simulation, we find an interesting phenomenon. Only adjusted by structural relaxation, the starting model of  $BP_4$  appears a certain degree of amorphization (Fig. 2b). That is to say losing long-range order for the component of  $BP_4$  is quite easy. We define this relaxed  $BP_4$  as natural amorphous  $BP_4$ , abbreviated as  $n\text{-}BP_4$ . As can be seen from Fig. 2b, compared with ZB  $BP_4$  (Fig. 2a), the local optimized  $BP_4$  ( $n\text{-}BP_4$ ) exhibits the disordered structure. The exact degree of order for ZB  $BP_4$  and  $n\text{-}BP_4$  is demarcated by the pair correlation functions (PCF) in Fig. 3a. We can see that, when the atomic distance is over 7 Å, there is no any visible feature of order degree for  $n\text{-}BP_4$ , which indicates that  $n\text{-}BP_4$  is indeed disordered. This character is consistent with our PACVD experiment that  $BP_4$  sample is indeed amorphous.

The above simulation just indicates the tendency of losing long-range order for  $BP_4$  but still record the short-range character from the starting ZB structure, hence we use M–Q method in MD to further research the more actual model of amorphous  $BP_4$ . The process of M–Q is as followed. First,  $BP_4$  is melt at 4000 K with 2000 steps (3 fs/step) MD simulation, which makes  $BP_4$  fully diffuse to eliminate the memory effect of original ZB structure. Second,  $BP_4$  is quenched to 300 K with 6000 steps (1 fs/step). Third, the geometry optimization is implemented to the quenched structure in order to amend the volume of  $BP_4$  during *NVT* MD. Fig. 2c shows the microscopic atomic picture of amorphous  $BP_4$  by M–Q method. From Fig. 2c, it can be seen that the present model of amorphous  $BP_4$  both loses the feature of long-range order and the character of original ZB structure. Fig. 3a also shows the PCF for  $BP_4$  with M–Q method (termed  $mq\text{-}BP_4$  thereof). It exhibits that, when the atomic distance is greater than 5 Å, there is also no any visible feature of order degree for  $mq\text{-}BP_4$ . After high-energy melting, the atoms of  $mq\text{-}BP_4$  are fully diffusive. Therefore, the local structure of amorphous  $mq\text{-}BP_4$  is quite different from that of  $n\text{-}BP_4$ . The most striking difference between  $mq\text{-}BP_4$  and  $n\text{-}BP_4$  is that the new peaks appear at 1.74 Å and the peak at 3.30 Å shifts to left, which means that the local neighbors of  $mq\text{-}BP_4$  become a slight closer. Between the short and middle atomic distance, 2–7 Å, the  $n\text{-}BP_4$  maintains a large degree of crystal structure but  $mq\text{-}BP_4$  does not. As  $mq\text{-}BP_4$  does not have the memory effect of original ZB structure, it is a typical representation for amorphous  $BP_4$ . We take  $mq\text{-}BP_4$  for further analysis.

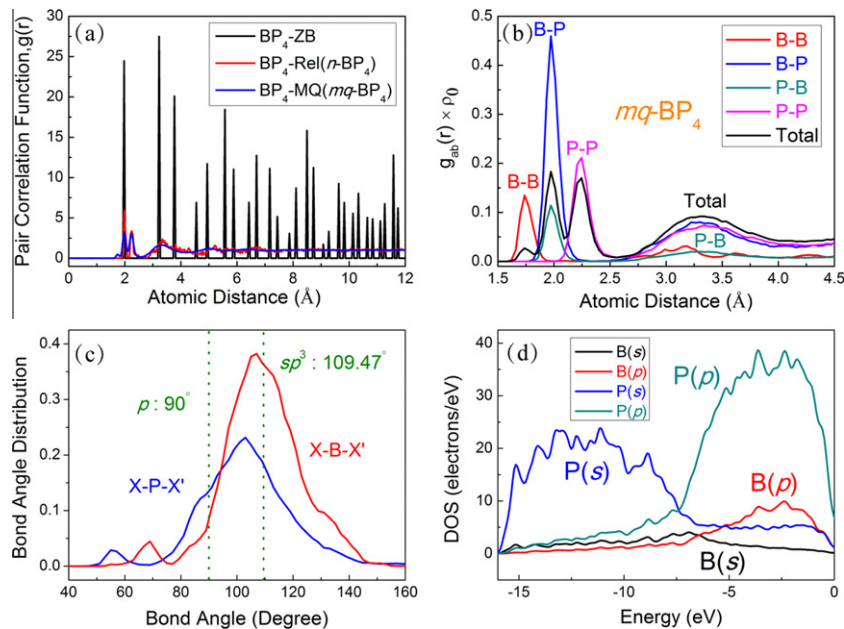
In order to detailed demarcate the PCF for different atomic pairs and their weight, we separate the total PCF of  $mq\text{-}BP_4$  to partial PCF ( $g_{ab}$ ) and then multiply  $g_{ab}$  by their corresponding atomic density, as shown in Fig. 3b. Three peaks in the nearest neighbors, B–B bonds at 1.74 Å, B–P bonds at 1.97 Å and P–P bonds at 2.25 Å, are easily distinguished, which correspond to the sum of the covalent radius of 1.70 Å (B–B), 1.85 Å (B–P) and 2.00 Å (P–P). Based on the partial PCFs, we get the coordination number (CN) for B and P element, respectively. For B atom total CN(B) of 3.96 includes 0.72 of B–B and 3.24 of B–P whereas for P atom total CN(P) of 3.24 includes 0.81 of P–B and 2.43 of P–P. Fig. 3c is the distribution function of bond angle for B and P atom in  $mq\text{-}BP_4$ . The bond angles of B atom maintain the  $sp^3$  hybrid bonding characteristics of the ideal ZB  $BP_4$  with  $109.47^\circ$ . However, the bond angles of P atom significantly deviate from the original  $sp^3$  feature and also a shoulder around  $90^\circ$  can be observed. This phenomenon is consistent with the situation of coordination. For B atom, the most coordinated atoms are still P atoms as the case in ZB  $BP_4$ , so  $sp^3$  hybrid bonding is still a main effect. For P atom, the coordination number of P is obviously larger than that of B, so the hybrid effect becomes weaker. Fig. 3d further shows the element- and orbital-dependent density of state (DOS) for  $mq\text{-}BP_4$ . We can see that the obvious separation of *s*- and *p*-orbital electronic states for P element is in contrast to the case for B element. The bond angle of around  $90^\circ$  from P element indeed reflect the enhancement of *p*-orbital states near valence band maximum. Fig. 2d further highlights the  $90^\circ$  square ring in amorphous  $mq\text{-}BP_4$ . These deformed or distorted square rings are mainly composed of P atoms. This is a direct proof of existence of *p*-orbital bonding for P atoms. Therefore, P atoms should both have original  $sp^3$  and new *p*-orbital bonding characters in amorphous  $BP_4$ .

#### 4. Conclusions

First-principles MD calculations and PACVD experiments reveal the atomic picture and electronic structure of amorphous  $BP_4$ , which have not been addressed before. Upon completion of the work, we learn that losing long-range order for  $BP_4$  is quite easy, as reflected by  $n\text{-}BP_4$ . Through M–Q technique in MD,  $mq\text{-}BP_4$



**Fig. 2.** (a) Original atom model of BP<sub>4</sub> based on zincblende structure; (b) atomic model of BP<sub>4</sub> after optimization from (a); (c) atomic model of BP<sub>4</sub> by melting–quenching (M–Q) method and (d) square ring close to 90° (yellow color) in atomic model of BP<sub>4</sub> by M–Q method. P and B atoms are marked by purple and brown ball, respectively. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)



**Fig. 3.** (a) Pair correlation functions of zincblende BP<sub>4</sub> (BP<sub>4</sub>-ZB), natural amorphous BP<sub>4</sub> (n-BP<sub>4</sub>) and BP<sub>4</sub> by melting–quenching method (mq-BP<sub>4</sub>); (b) partial pair correlation functions of mq-BP<sub>4</sub> for different atom pairs multiplied by the corresponding atomic density; (c) atomic bond angle distribution function of mq-BP<sub>4</sub>, the blue line is for P atoms and the red line is for B atoms; (d) the element- and orbital-dependent density of state (DOS) for mq-BP<sub>4</sub>. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

model, a typical representation for amorphous BP<sub>4</sub> is obtained. In mq-BP<sub>4</sub>, B atoms retain  $sp^3$  electronic bonding while P atoms have coexistence of  $sp^3$  and  $p$  bonding. The cohesive energy reflecting

bond strength of material is  $-5.32$  eV/atom for mq-BP<sub>4</sub> and  $-3.37$  eV/atom for  $c$ -ZnS. This is consistent with the fact that amorphous BP<sub>4</sub> can be employed to protect  $c$ -ZnS.

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